



Corrosion Protection of Steel Parts in Aircraft by Zn-Mn Alloy Electrodeposits

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Abstract: Electrodeposited Zn-Fe, Zn-Co and Zn-Ni alloy coatings are widely used for corrosion protection of steel nowadays. Several authors have reported that in aggressive environments, such as atmosphere containing SO₂ and chloride compounds, Zn-Mn alloy coatings can have even better corrosion resistance. Besides, these alloys possess high hardness and they can easily be painted. Good characteristics of Zn-Mn alloys suggest their possible application in protection of steel parts, especially in automobile and aircrafts. In the last ten years, three new types of electrolytic solutions for electrodeposition of Zn-Mn alloys were proposed, namely acidic chloride and sulphate and alkaline pyrophosphate solution.

The scope of this work was to compare the corrosion protection ability of Zn-Mn alloy coatings electrodeposited under various parameters, such as type of electrolytic solution, concentration of metal ions in solution and deposition current density. Recording of corrosion potential and salt spray chamber measurements were used to investigate the corrosion stability of Zn-Mn coatings. It was concluded that Zn-Mn alloy had better corrosion resistance than pure Zn, due to the formation of compact protective-passive layer in chloride containing media. Among various Zn-Mn electrodeposits, the best corrosion behavior was obtained for the ones deposited from chloride solution, at current densities between 3 and 8 A dm⁻².

Keywords: Zn-Mn alloys, corrosion, salt spray chamber, steel protection

1. Introduction

The main demands for materials used in aircraft manufacture are to be corrosion and heat resistant, light, strong materials that are easily machined, assembled and repaired. Currently, materials used for this purpose are aluminum alloys, titanium and composites [1]. Stainless or coated steel are used for aircraft parts with special mechanical and electrical usage, such as turning parts, precision casting parts, engine parts, transmission components, fittings, landing gear trunnions etc [1, 2].

Electrodeposition of sacrificial coatings of zinc and its alloys, particularly Zn-Fe, Zn-Co and Zn-Ni, is a widely used method that provides corrosion protection of steel parts [3]. However,

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for corrosion protection of steel under severe atmospheric conditions, in particular in the presence of SO₂ and chloride compounds, various alternative materials are being investigated. Among those materials are the zinc-manganese alloys, which have been reported to show better corrosion resistance properties than other zinc alloys. Zn–Mn system seems to display both improved corrosion resistance and hardness and good environmental compatibility, so it is acceptable for protection of aircraft steel parts [4, 5].

Zn-Mn coatings are less thermodynamically noble than Zn, but they show a passivating behavior in various environments due to the formation of insoluble compounds on the corroding surface. Depending on aggressive media, various compounds were found in passive layer, such as oxides MnO, Mn_{0.98}O₂, Mn₅O₈ [6] and γ -Mn₂O₃ [7], or basic salts like Zn₄(OH)₆SO₄·xH₂O and Zn₅(OH)₈Cl₂·H₂O [8].

Corrosion stability of electrodeposited alloy coating depends on its surface morphology, chemical and phase composition. These properties are strongly influenced by the type of electrolytic solution used for electrodeposition and plating parameters, such as deposition current density, temperature, stirring of the solution, etc. [9]. The scope of this work was to compare the corrosion protection of Zn-Mn alloy coatings electrodeposited at various deposition current densities. Three different electrolytic solutions were used: slightly acidic chloride and sulphate solution and alkaline pyrophosphate solution. The change of corrosion potential with time of immersion in NaCl solution and salt spray chamber measurements were used to investigate the corrosion stability of Zn-Mn coatings. The surface coatings roughness was also measured, in order to examine the surface morphology of the coatings.

2. Experimental Work

Electrodeposition of Zn-Mn alloys was performed on steel panels galvanostatically, at various current densities, of 1-12 A dm⁻². Deposition times were managed to obtain layers of typically 10 μ m thickness. The composition of the plating baths used is presented in Table 1. All solutions were made from analytical grade chemicals (Sigma Aldrich) and double distilled water.

The electrodeposition process was carried out in a classical three-electrode glass cell, at the temperature of 20 \pm 2 °C in aerated conditions, without stirring. Working electrodes were steel panels. Prior to each experiment, the steel surfaces were polished successively with emery papers of the following grades: 600, 1000, 1200 and then degreased in a saturated solution of NaOH in ethanol, pickled with a 2 mol dm⁻³ HCl solution for 30 s and finally rinsed with distilled water. Cylindrical Zn foil (high purity zinc, 99.9 mass%), placed close to the cell walls, was used as a counter electrode, providing good current distribution in the cell, since working electrode was placed in the middle of the cell. The reference electrode was a saturated calomel electrode (SCE).

Table 1 Composition of plating baths

Chloride	$c / \text{mol dm}^{-3}$	Sulphate	$c / \text{mol dm}^{-3}$	Pyrophosphate	$c / \text{mol dm}^{-3}$
KCl	3.2	MnSO ₄ · H ₂ O	0.30	K ₄ P ₂ O ₇	1.0
H ₃ BO ₃	0.42	ZnSO ₄ · 7H ₂ O	0.30	ascorbic acid	0.03
ZnCl ₂	0.45	(NH ₄) ₂ SO ₄	0.49	MnCl ₂ · 4H ₂ O	0.05
MnCl ₂ · 4H ₂ O	0.45			ZnCl ₂	0.05

The corrosion behavior of the Zn-Mn coatings was first studied by neutral salt spray tests, according to ASTM B-117-07a standard, using a Q-panel chamber (Mod. CCT-1100). The tests were performed using a 0.5 mol dm^{-3} NaCl solution with a pH in the range of 6.5–7.2. For these tests, the coatings were deposited on steel panel (30 x 50 mm). The exposure time was recorded at which red corrosion was first observed; such corrosion is associated with the oxidation of the steel substrate.

The deposit thickness was $\sim 10 \mu\text{m}$, measured by Coating Thickness Measuring Instrument DUALSCOPE MPOR. Chemical composition of electrodeposited Zn-Mn alloys was determined by atomic absorption spectrophotometry, using atomic spectrophotometer PYE Unicam SP9, Philips. The $10 \mu\text{m}$ thick pure zinc coatings were also tested in salt spray test, for comparison.

In addition, the corrosion behavior of Zn-Mn coatings was studied by following the corrosion potential (E_{corr}) during exposure to aerated 0.5 mol dm^{-3} NaCl solution, until the E_{corr} of the steel substrate was reached. Steel panels ($20 \times 20 \text{ mm}$) were used as substrate for alloy coating deposition. The deposit thickness was also $10 \mu\text{m}$. The corrosion potential was recorded by a potentiostat / galvanostat ZRA Reference 600, Gamry Instrument.

Roughness of the Zn-Mn deposits was measured using TR200 Surface Roughness Tester. Results were recorded by computer attached to the surface tester, using software TR200 Time Data View.

3. Results and Discussion

3.1 Salt Spray Chamber

To evaluate the protective capacity of the Zn-Mn coatings on the steel substrate, salt spray tests were performed. The results obtained are presented in Table 2 and they show that most of the Zn-Mn samples lasted longer before the appearance of red corrosion, as compared to pure Zn samples. Manganese contents in these alloys, determined by atomic absorption spectrophotometry, are presented also in Table 2. These results clearly prove that plating baths investigated in this work can be successfully used for electrodeposition of Zn-Mn alloys, in order to increase the corrosion stability of pure Zn coatings.

However, there is a significant difference in the salt spray test behavior between Zn-Mn coatings deposited under various conditions. It can be noticed that coatings deposited at the highest deposition current densities ($8 \text{ i } 12 \text{ A dm}^{-2}$) withstood considerably shorter exposure time before the onset of red corrosion, compared to samples deposited at lower current densities. After 96 hours of exposure in the salt spray chamber, minimal damage was observed at samples deposited from chloride solution, with metal ions ratio of $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 4 : 1$, at current density of 3 A dm^{-2} .

3.2 Corrosion Properties of Zn–Mn Alloys in Aqueous Solution

In order to determine the corrosion stability in chloride media, the plated specimens were immersed in 0.5 mol dm^{-3} NaCl solution and the E_{corr} was measured daily. Fig. 1 shows the time dependence of E_{corr} for steel plated by Zn-Mn alloys deposited from three different plating baths. The E_{corr} of bare steel surface in 0.5 mol dm^{-3} NaCl was -640 mV vs. SCE and it is marked with a line in Figure 1. The potential of Zn-Mn alloys is more negative than E_{corr} of steel base, so Zn-Mn deposits offer sacrificial cathodic protection. The E_{corr} values of steel

modified by Zn-Mn alloys increase positively with time of immersion and reach the steel E_{corr} , which represents the deposit loss and start of the corrosion process at the substrate.

The time of reaching the steel E_{corr} for different specimens is presented in Table 3. Similarly to the salt spray chamber results, the best corrosion stability of Zn-Mn coatings is obtained by deposition from chloride solution. The coatings deposited at lower current densities showed higher stability.

The time for the red rust appearance, i.e. for reaching the E_{corr} of the steel substrate, is determined by two factors, thermodynamic and kinetic ones. The first factor refers to the initial E_{corr} value of the alloy. As a rule, alloys with more positive initial E_{corr} value are thermodynamically more stable and the change of their E_{corr} toward the steel E_{corr} is slower [3]. As Figure 1 depicts, the E_{corr} of Zn-Mn alloys obtained at different parameters, slightly differs initially, which can be attributed to the alloy phase difference and the difference in the chemical composition. Kinetically factor which determines corrosion stability of the Zn-Mn alloy specimens refers to the formation of protective corrosion product layer on the corroding surface. Figure 1 reveals that after 24 hours of immersion in 0.5 mol dm^{-3} NaCl solution, all specimens have very close E_{corr} values. After this point, E_{corr} values reach a plateau for all investigated samples. This is strong evidence that protective layer is formed on corroding samples [10]. The abrupt change of the immersion potential in positive direction, for prolonged immersion times, indicates that contact of the electrolyte with Fe-substrate is made through possible pinholes of the Zn-Mn electrodeposits, which means that the protective layer on the Zn-Mn coating is disturbed and degradation of the coating and the substrate occurs.

Table 2 The exposure time at which oxidation of the steel substrate (red corrosion) was first observed in the salt spray test.

Plating solution	Deposition current density / A dm^{-2}	Mn content in the coating (mass. %)	Time / h
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	3	1.5	72
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	8	6.3	24
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	12	9.9	8
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:1$	3	0.5	72
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:1$	8	1.1	8
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:2$	3	0.4	72
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:2$	8	0.9	8
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 4:1$	3	0.7	72
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 4:1$	8	6.2	24
Sulphate	3	1.2	72
Sulphate	5	3.5	24
Pyrophosphate	3	4.7	24
Pyrophosphate	8	10.1	8
pure Zn	3	0	24
pure Zn	8	0	24
pure Zn	11	0	24

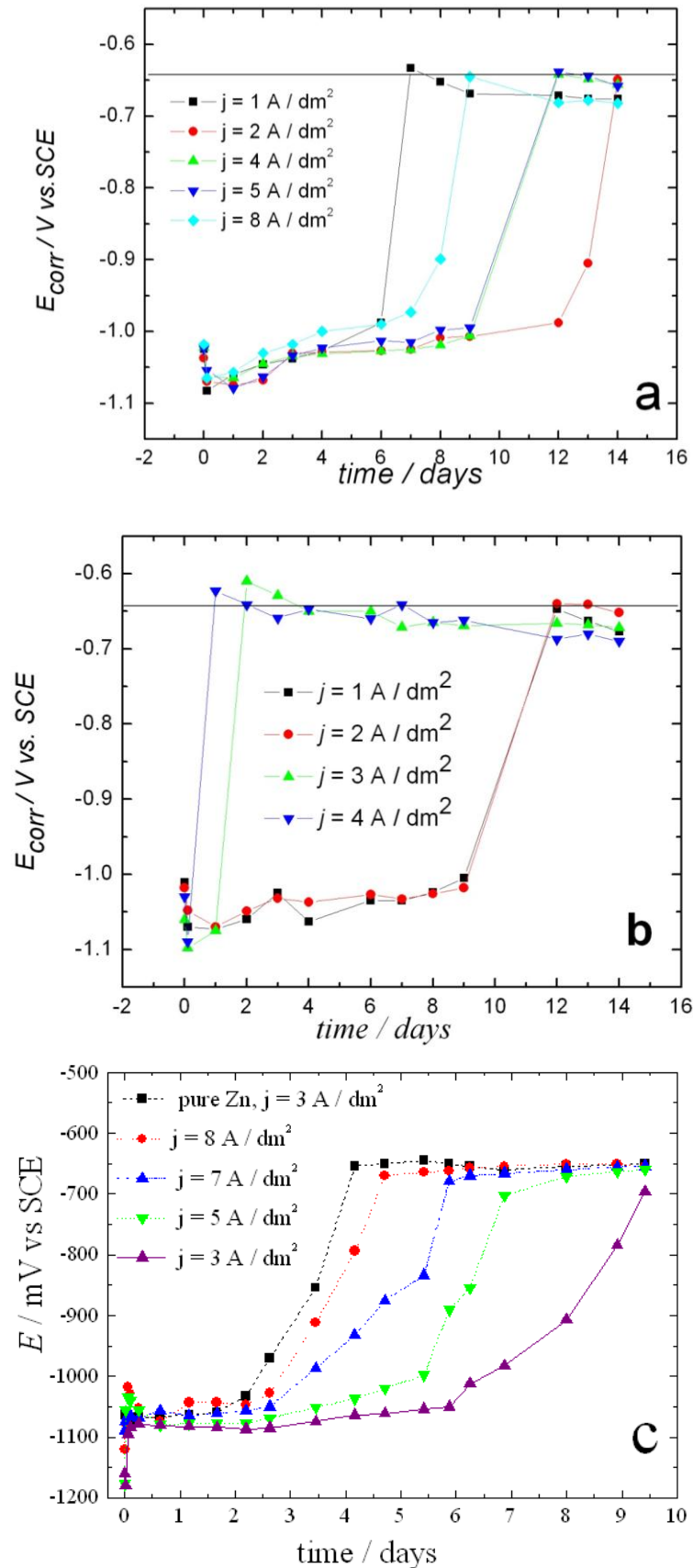


Fig. 1. The dependence of E_{corr} with time of immersion in 0.5 mol dm⁻³ NaCl solution, for pure Zn and Zn-Mn alloys deposited from: a) chloride solution, [Mn²⁺]:[Zn²⁺] = 2:1; b) sulphate solution, [Mn²⁺]:[Zn²⁺] = 1:1 and c) pyrophosphate solution, [Mn²⁺]:[Zn²⁺] = 1:1

Table 3 The exposure time after which E_{corr} of the steel substrate was reached in the 0.5 mol dm^{-3} NaCl solution

Solution	[Mn ²⁺] : [Zn ²⁺]	$j / \text{A dm}^{-2}$	Time / days
Chloride	1 : 1	1	9
		2	17
		7	14
	1 : 2	1	9
		2	9
		4	12
		11	12
	2 : 1	1	7
		2	14
		4	11
		5	11
		8	9
	4 : 1	3	12
		5	12
		7	9
		9	9
Sulphate	1 : 1	1	12
		2	12
		3	4
		4	2
Pyrophosphate	1 : 1	2	7
		3	6
		8	6
		12	5

3.3 Surface Roughness

In order to examine surface morphology of electrodeposited Zn-Mn alloy coatings, roughness of the surfaces was measured at the length of 2.5 mm, as it is shown in Figure 2. The parameters obtained were as follows: R_a – average roughness, i.e. the arithmetical average deviation of the surface profile (equation 1) and R_z - the maximum height of irregularities (equation 2).

$$R_a = \frac{1}{n} \sum_{i=1}^n |y_i| \quad (1)$$

$$R_z = R_v + R_d \quad (2)$$

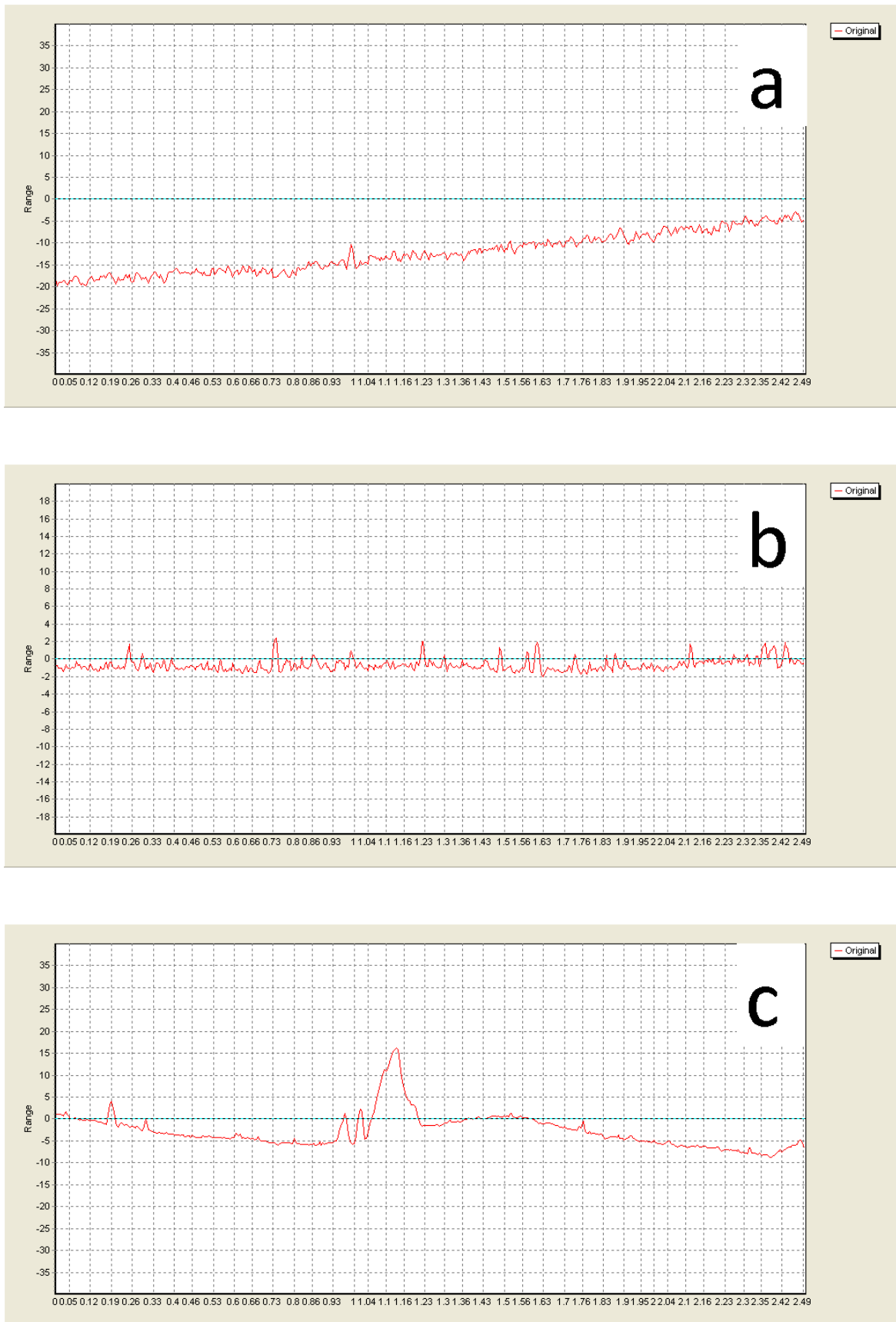


Figure 2 Profile lines for Zn-Mn alloy surfaces deposited at 3 A dm^{-2} from a) chloride; b) sulphate and c) pyrophosphate solution.

Surface roughness of the electrodeposited coating gives information about homogeneity of the coating, size distribution of crystalline grains, and current density distribution on the working electrode [11]. Low values of R_a and R_z parameters for one coating indicate that the coating is homogeneous, non-porous and defect free. So, compact protective layer of corrosion product has better chance to form on the surface with low roughness [11].

Table 4 depicts calculated roughness parameters. Some conclusions about the course of the change in the surface roughness of Zn-Mn coatings with change of deposition parameters can be made based on the results shown in table 4 and Figure 2. First, independently on plating solution used, the increase in deposition current density results with the increase in roughness of deposits. Second, when plating solutions are compared, it can be seen that coatings deposited from the sulphate solution possess the lowest roughness, and coatings deposited from chloride solution are just slightly coarser. Contrary, deposits obtained from alkaline pyrophosphate solution possess significantly higher roughness.

As a conclusion, roughness results are in correlation with corrosion investigations to some extent. Smoother specimens have shown higher corrosion resistance and vice versa. It is strong evidence that the most important factor in corrosion stability of Zn-Mn alloys is the stability of corrosion product layer on their surface.

Table 4. Roughness parameters for Zn-Mn coatings deposited under various conditions

Solution	$j / \text{A dm}^{-2}$	$R_a / \mu\text{m}$	$R_z / \mu\text{m}$
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:2$	3	0,748	6,46
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:1$	3	0,528	3,42
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	3	0,658	5,26
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 4:1$	3	0,904	15,97
Sulphate	3	0,42	3,88
Pyrophosphate	3	2,754	23,39
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:2$	8	1,652	13,03
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:1$	8	0,847	6,94
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	8	0,874	10,81
Pyrophosphate	8	2,989	30,84
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 1:1$	12	1,274	14,3
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 2:1$	12	1,159	9,659
chloride, $[\text{Mn}^{2+}] : [\text{Zn}^{2+}] = 4:1$	12	1,96	18,02

4. Conclusion

Although main aircraft materials are light metals like titanium and aluminum alloys, the use of steel is unavoidable for some parts of the aircraft. Zn-Mn alloy coatings are recognized as successful possible method to protect steel surface from corrosion, due to their stability in aggressive atmosphere and high hardness.

Zn-Mn coatings can be electrodeposited from various plating solutions. In this work, three plating solutions were used and corrosion stability of obtained deposits was examined. It was shown that coatings obtained from chloride solutions possessed significantly higher corrosion stability in comparison with sulphate and pyrophosphate solutions. The best corrosion performance possessed coatings deposited at current densities in the range of 2 – 5 A dm⁻². Surface morphology of the deposits was examined through the roughness measurements. It was concluded that coatings obtained from slightly acidic solutions are significantly smoother as compared to the coatings from alkaline pyrophosphate solution. Also, it was shown that smoother coatings possess better corrosion stability in chloride media.

5. References

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